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Thermal characterisation of metal stearate lubricant mixtures for polymer compounding applications

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Highlights

- Thermal characterization of binary blends of metal stearates was performed.
- Blends melted at lower temperatures and featured lower melt viscosities.
- The metal stearates co-crystallised instead of forming eutectic mixtures.
- Metal stearate blends imparted lower melt viscosity to chalk-filled polyethylene.
- Particle agglomeration was reduced and particle dispersion was improved.

Abstract

Metal stearate blends are potential lubricants and filler dispersants in polymer compounding. Therefore, the thermal phase behaviour of binary blends of calcium stearate and magnesium stearate with zinc stearate was investigated. Cooling curve studies indicated lower melting temperatures in both systems. X-ray diffraction studies revealed that this was due to the formation of solid solutions of variable composition rather than to eutectic formation. Rheology measurements showed that true fluidification only happened well above the apparent melting points established from cooling curves. Calcium carbonate filled polyethylene containing metal stearate mixtures showed lower melt viscosities than were achievable using only calcium- or magnesium stearate as the lubricant. These results suggest that metal stearate blends, in combination with a judiciously selected wax component, may offer advantages in lowering compound melt viscosity and facilitating better particle dispersion.

Graphical abstract



Keywords: metal stearate; lubricant; dispersant; thermal analysis; filler

1 INTRODUCTION

The metallic stearates, zinc stearate (ZnSt), calcium stearate (CaSt) and magnesium stearate (MgSt), serve as multipurpose polymer additives [1]. They are used as antacids in polyolefins [1], as co-stabilisers in PVC formulations [2, 3], as general mould release agents [4] and also as lubricants [5, 6]. Compared to other metallic stearates, zinc stearate appears to be the most effective lubricant for many applications including polymers [3, 4].

Metal stearates are also effective dispersants for inorganic fillers such as calcium carbonate. This is probably due, in part, to filler surface modification that renders it compatible with the hydrophobic polymer, e.g. linear low-density polyethylene (LLDPE). This facilitates the break-down of particle agglomerates and dispersion of individual particles in the polymer matrix [7]. This is a dynamic process that involves diffusion of the dispersants into the cavities of particle agglomerates. It reduces the shear forces required for disintegration and dispersion of the individual filler particles during the compounding process. A lubricant system with a lower melt temperature, i.e. a reduced fluidisation temperature could assist this process. Binary mixtures of compounds do show such reductions in melting behaviour, most commonly due to eutectic formation. Therefore, it was of interest to determine whether this would also apply for mixtures of metal stearates.

Combinations of calcium- and zinc stearates are used for heat stabilisation of PVC. The application of such mixtures as lubricants in filled polyolefin-based compounds has not yet

been reported. Perhaps such mixtures may provide improved lubricant and filler particle dispersion properties in highly filled polymer compounds and masterbatches. As a first step towards exploring this possibility, the properties of binary combinations of zinc stearate with either calcium- or magnesium stearate were investigated. The thermal phase behaviour and the transition to a liquid state of such blends were investigated as they bear relevance to their performance as filler particle dispersants and also as lubricants and viscosity reduction agents in polymer compounding applications.

2 MATERIALS AND METHODS

2.1 Materials

Technical grade samples of the metal soaps were procured. The magnesium stearate [CAS No. 557-04-0] (Product code: 415057) was obtained from Sigma-Aldrich. The calcium stearate grade SAK-CS [CAS no. 1592- 23-0] and zinc stearate grade SAK-ZS-TGF [CAS no. 557-05-1] were supplied by Sun Ace.

Sasol supplied a proprietary experimental Fischer-Tropsch wax (F-T wax) and linear lowdensity polyethylene (LLDPE) grade HM2420. The density of the latter was 924 kg·m³ and the melt index was 20 g/min @ at 190°C/2.16 kg. Safripol supplied HDPE Safrene M9255F. The density of this film blowing grade was 953 kg·m³ and the melt flow rate was 0.3 g/min @ 190°C/5 kg. Serina supplied stearic acid-coated calcium carbonate grade Ascom 50T. According to the supplier, the mean particle size was 2.2 μm.

2.2 Methods

2.2.1 Material preparation

The following calcium carbonate-filled LLDPE compound formulation was used in order to obtain a preliminary indication of the lubrication effect imparted by the metal stearates: LLDPE (39.3 wt.%); Calcium carbonate (30 wt.%); wax (23.0 wt.% Sasol F-T wax) and metal stearate (7.7 wt.%). The low melt viscosity combination of polyethylene with Sasol F-T wax was chosen as this made the filled compound more amenable for evaluation on the plate-plate rheometer. Compounding was performed on a ThermoFischer TSE 24 co-rotating twinscrew compounder (24 mm ϕ , 30 L/D). Based on past experience, the screw speed was set at 80 rpm and the temperature profile, from hopper to die, was 60/110/140/170/170/170/170/20 °C.

2.2.2 Film blowing

The HDPE films, with 10 % add-on of the prepared calcium carbonate masterbatches, were blown on a Collin BL 180/400 blown film unit. It comprised a 30 mm ϕ single screw extruder with L/D = 25. The extruder die had a diameter of 60 mm and featured a dual-lip cooling ring. The extruder was operated at a screw speed of 70 rpm. Stable film production was attained when profiles, die. the temperature from hopper to were set at 190/190/195/195/195/195/220/220 °C.

2.2.3 X-ray diffraction methodology

X-ray diffractograms were recorded on a Bruker D2 PHASER instrument using Cu K α radiation (λ =1.54060). The system was equipped with a LYNXEYE_XE-T detector with an up to 4.99° PSD opening. Samples were run as pressed powder pellets and scanned from 1° to 50° 20 at a rate of 0.02° 20 steps per second.

2.2.4 Differential scanning calorimetry (DSC)

DSC was performed on a Perkin Elmer Pyris 4000 Instrument. Samples of the metal stearates, weighing 8 ± 1 mg were placed in aluminium pans covered with a lid with a pinhole. The temperature was scanned from 20 °C to 200 °C at a rate of 10 °C·min⁻¹ with nitrogen flowing at a rate of 20 mL·min⁻¹.

2.2.5 Thermogravimetric analysis (TGA)

TGA was performed on a TA Instruments SDT-Q600 Simultaneous TGA/DSC instrument. Samples weighing ca. 8 ± 1 mg were placed in 70 µL alumina cups. The temperature was scanned from 20 °C to 700 °C at a rate of 10 °C·min⁻¹ with nitrogen flowing at a rate of 20 mL·min⁻¹.

2.2.6 Energy Dispersive Spectroscopy (EDS)

A Supra 55VP Field Emission scanning electron microscope fitted with an Oxford energy dispersive spectroscopy (EDS) system was used to visualise the size and distribution of CaCO₃ aggregates within intact films. The analysis of carbon coated polymer films was performed side-on, at 25 kV and collecting 300k counts for quantitative EDS spectra, and over 500k for each X-ray map. The choice of such high accelerating voltage aimed to probe the polymer film sub-surface to a depth in excess of 4.5 µm [8]. Since the calcium-containing

particles were roughly 2 μ m in diameter, this depth would have revealed up to about three layers of particles in the 5 μ m depth interrogated by the SEM beam.

Elemental map images were opened in ImageJ, calibrated by drawing a line over the scale bar and entering 50 in the Analyze>Set Scale>Known Distance dialogue box). Images were segmented (Image>Adjust>Color Segmentation) using the default setting in the Color Thresholding RGB (ticking Red channel only and "Dark Background" options). The threshold areas were "Selected" in the Threshold Colour Panel, the image was converted to a binary image (Process>Binary>Make Binary), the "Fill holes" algorithm was applied (Process>Binary>Fill Holes), followed by "Analyze Particles" (Analyze>Analyze Particles), which generated a set of numbered outlines. From this the Feret diameter was extracted as the measure of the size of the CaCO₃ deposits. Results were exported to Excel and subjected to an F-test (to explore if they had equal or unequal variances) and by a t-test to determine whether the sample means were statistically different from each other.

2.2.7 Rheology

The rheological experiments were performed on an Anton Paar MCR301 rheometer with a Peltier heating attachment. The experiments were performed with a 50 mm parallel plate configuration and a gap setting of 1 mm. The metal stearate samples were melted at 180 °C and squeezed to a gap setting of 1 mm. The sample was pre-sheared for 1 minute at a shear-rate of 5 s⁻¹ followed by 1 minute of rest. The temperature was then scanned from 180 °C to 60 °C and then back up to 180 °C at a scan rate of 5 °C·min⁻¹. This was done in oscillation mode at a frequency of 10 rad·s⁻¹ and a strain of 0.5 %. The normal force was controlled at 0 N.

Rheological measurements on the calcium carbonate-filled polyethylene samples were obtained with a frequency sweep from $0.2 \text{ rad} \cdot \text{s}^{-1}$ to 200 rad $\cdot \text{s}^{-1}$. The sample oscillation was specified at a fixed strain of 0.05%. The temperature was controlled at 170 °C.

2.2.8 Optical microscopy

Microscopy experiments were conducted on a Leica DM2500M optical microscope connected to a Leica DFC420 digital camera. The optical micrographs were obtained under polarised light with a 1 λ retarder plate. The microscope was operated with a Linkam

Scientific CSS450 heating stage. The samples were crushed between a glass slide and the glass surface of the stage. The temperature was scanned from 30 °C to 180 °C at a heating rate of 10 °C·min⁻¹ after which the molten material was squeezed to remove most of the trapped air bubbles. The temperature was then reduced from 180 °C to 30 °C at a cooling rate of 10 °C·min⁻¹ in order to observe the crystallization. The samples were held isothermally for 1 min at 10 °C temperature halts during both the heating and cooling cycles to allow for temperature equilibration. Occasionally this interval was changed to 5 °C based on observations made during the experiments. Micrographs were captured at the intervals after 1 minute of rest at 2.5×, 10× and 20× magnification.

2.2.9 Cooling Curves

Solid lubricant samples weighing 5.0 g were used for the cooling curve experiments. These samples were well mixed and mixtures were prepared using 10 wt.% increments. The solid material was added to a test tube and heated in a silicone oil bath on a Dathan Scientific MSH 20D hotplate stirrer. The temperature was adjusted until all the solids were molten. A temperature probe, attached to a Hairuis SSN-61 data logger, was placed in the sample. The temperature was logged every 2 seconds as the sample was allowed to freely cool down to 50 °C. Undercooling and temperature rebound was monitored to identify the apparent melting temperature.

3 RESULTS AND DISCUSSION

The commercial metal soaps, used presently, are actually complex mixtures of salts of several different fatty acids. They consist mainly of stearic acid and palmitic acid with the total amount of these acids being at least 90 % [9]. Unfortunately, the suppliers did not disclose the composition of the actual products used in this investigation. For calculation purposes, it was assumed that the samples all contained 65 wt.% metal stearate and 35 wt.% of the metal palmitate.

The calcium and magnesium stearates may exist in multiple crystal forms associated with different degrees of hydration. Figure 1 shows the mass loss curves of the neat metal stearates obtained in a nitrogen atmosphere. The mass loss recorded at a temperature of 150 °C provides an indication of the moisture content or the amount of water of hydration present. The mass loss of 0.2 wt.% for the zinc stearate probably represents adsorbed moisture. The

2.9 wt.% mass loss found for calcium stearate is close to that expected for the monohydrate (3.0 wt.%). The recorded mass loss for the magnesium stearate (4.1 wt.%) suggests that the neat product was a mixture of the mono hydrate and the dihydrate. The expected mass loss values for those two forms are 3.1 wt.% and 5.9 wt.% respectively. Figure 1 also reveals that the thermal stability of the three soaps decreased in the order of the metal ion present according to the series calcium > zinc > magnesium stearate.



Figure 1. Thermogravimetric analysis of the neat stearates in a nitrogen atmosphere

Figure 2 reports the X-ray diffraction patterns of the metal stearates recorded at ambient temperature. It shows diffractograms obtained for the as-received powders and also after a heat-treatment at 180 °C, i.e. an exposure to a temperature higher than the melting points. The diffraction patterns of the calcium and zinc stearates are approximately the same for the neat and the heated forms. The only difference is that the reflections of the latter forms are slightly broader, indicative of reduced crystalline perfection. The diffractogram for the neat magnesium stearate is consistent with the dihydrate form. However, the diffraction pattern of the heat-treated magnesium stearate differed significantly from that of the neat sample.



Figure 2. X-ray diffractograms recorded for the neat metal stearates as received and after a heat treatment at 180 °C (indicated by (m) for melted samples).



Figure 3. X-ray diffractograms for binary mixtures of zinc stearate with (a) calcium stearate, and (b) magnesium stearate. All the diffractograms were recorded after the mixtures were mixed at 180 °C and allowed to cool down to ambient temperature.

Figure 3(a) shows the XRD diffractograms for the binary calcium/zinc stearate blends. From 0 wt.% to 80 wt.% zinc stearate the diffraction patterns are similar to those observed for the calcium stearate. This suggests that the two metal stearates co-crystalize. As revealed in Figure 4, the d-spacing remained constant at 4.82 ± 0.06 nm. This is consistent with the zinc

stearate being progressively incorporated into a calcium stearate-defined lattice. The formation of this multicomponent crystal phase is doubtless facilitated by the smaller ionic radius of Zn^{2+} ions (74 pm) compared to the radius of Ca^{2+} ions (114 pm). This makes it easy for a zinc ion to replace a calcium ion in the crystal lattice.



Figure 4. Variation of the d-spacings corresponding to the migrating reflections, at angular range 1° to $3^{\circ} 2\theta$, with zinc stearate content of the binary metal stearate blend. The labels (a) and (b) refer to d-spacings corresponding to the first two reflections observed in the magnesium stearate-zinc stearate blends.

Figure 3(b) shows the XRD patterns for the binary magnesium/zinc stearate blends. Figure 4 shows a plot of the d-spacings as a function of blend composition. The d-spacing of the neat zinc stearate is 4.29 nm. As the magnesium content was increased to 40 wt.% the d-spacing increased approximately linearly. Beyond 50 wt.% magnesium stearate content, the angle corresponding to this reflection remained approximately constant corresponding to a d-spacing of 5.10 ± 0.10 nm. This is lower than the d-spacing of the neat hydrated magnesium stearate, i.e. 5.49 nm. However, it does mean that the presence of even minor amounts of zinc stearate assists the formation of a crystalline phase similar to that of a different dehydrated magnesium stearate. Coming from the opposite side of the composition range, one observes that the main reflection of the dehydrated magnesium stearate is also observed in blends

containing up to 50 wt.% zinc stearate. The d-spacing corresponding to this reflection is 3.15 ± 0.04 nm. This means that the XRD data indicate that at least two different phases are present below a zinc stearate content of about 50 wt.%. Furthermore, the XRD results are also consistent with co-crystallization occurring at higher zinc stearate content.



Figure 5. Differential scanning calorimetry calcium/zinc stearate blends showing the variation of the melting behaviour with composition.

Figure 5 shows the DSC heating endotherms found, during heating scans, for previously melted and solidified calcium stearate, zinc stearate and selected blends. Both the calcium and the zinc stearate show well-defined, single-event endotherm peaks. The peak temperatures were 128.3 °C and 124.5 °C for the neat calcium- and the neat zinc stearate samples respectively. In all the blend samples, the endotherms peaks are shifted to significantly lower temperatures. The sample containing 20 wt.% zinc stearate also featured a single endotherm peak but a shoulder is noticeable at the lower temperature side. In the blend containing 40 wt.% zinc stearate, this endothermic event developed into a separate peak located at 101.1 °C. Multiple endotherms event are indicated by the peaks and shoulders visible in the samples containing 60 wt.% and 80 wt.% zinc stearate. The presence of multiple endothermic events does not necessarily invalidate the deductions made on the basis of the XRD results. This is because rotator phase transitions and mesophase formation are also endothermic events [10].

Figure 6 shows the DSC endotherms observed on heating samples of the magnesium-zinc stearate binary system. Compared to the other two metal soaps, magnesium stearate (MgSt) shows highly complex thermal phase behaviour [9, 10]. According to Haware, Vinjamuri, Sarkar, Stefik and Stagner [10], the thermal transitions observed during the first heating cycle of hydrated forms are irreversible. Thereafter, complex cool-heat-cool-heat DSC thermograms are reproducible and exhibit up to six reversible exothermic-endothermic conjugate pairs [10]. These include five distinct mesophases stable at different temperatures with one phase persisting up to 250 °C. Curiously, molecular ordering was maintained beyond 276 °C implying that magnesium stearate does not undergo a simple melting phenomenon [10]. Similar observations were made in the present investigation.



Figure 6. Differential scanning calorimetry magnesium stearate – zinc stearate blends showing the variation in the nature of the endothermic events with composition

Unlike the zinc stearate, the melting thermogram for the magnesium stearate considered presently, featured multiple endotherms. The first endotherm of the anhydrous form is probably due to a transformation into a rotator structure while those occurring at higher temperatures correspond to the melting of this phase and finally the transformations into an inverse micelle phase [11]. Three endotherm peaks, shifted to lower temperatures, are also observed when just 10 wt.% zinc stearate is present. However, the sample containing 40 wt.% zinc stearate featured only two endotherms. Their peak temperature positions agreed well

with the first and third endotherm of the sample with 10 wt.% zinc stearate. Samples containing even more zinc stearate featured multiple endothermic events as distinct peaks or as shoulders.



Figure 7. Estimating the temperature associated with the onset of melt fluidity from plots of the complex viscosity: viscosity: Blends of zinc stearate with (a) calcium stearate and (b) with magnesium stearate. The measurements reflect data obtained during a heating scan and the curves were shifted vertically to a common plateau value of 80 kPa·s in the temperature range 70 °C to 90 °C.

Figure 7 shows shifted complex viscosity curves for neat metal stearate and their binary blends. The low-temperature plateau values reflect the rheological behaviour of the solid state. Unfortunately, the complex viscosity values in this range showed considerable scatter whereas the onset temperatures for melt fluidity were reproducible. In order to make the latter easier to perceive, all the curves were shifted vertically to a common plateau value 80 kPa·s. Figure 7(a) shows the data for the calcium-zinc stearate binary system. All the blends became fluid at temperatures lower than the two parent compounds, i.e. 120 °C and 141 °C for zinc-and calcium stearate respectively. The lowest fluidification temperature (96.5 °C) was found for the blends containing 60 wt.% zinc stearate. This correlates well with the DSC results curves which also show that this mixture has the lowest observed melting endotherm.

Figure 7(b) shows the corresponding curves for the magnesium-zinc stearate binary. This system showed more complicated behaviour. Most blends liquified at temperatures between 120 °C and 151 °C, i.e. the fluidification temperatures associated with the parent zinc- and

magnesium stearate respectively. The exceptions were the blends containing 35 wt.% and 40 wt.% zinc stearate. These two mixtures had fluidification onset temperatures lower than that of the neat zinc stearate. Actually, diluting the magnesium stearate with up to 30 wt.% zinc stearate did not change the fluidification temperature materially. Similarly, addition of up to 20 wt.% calcium stearate did not change the fluidification temperature (106.5 °C) was obtained with a blend containing 65 wt.% zinc stearate.

Figure 8 compares the fluidification onset temperatures to the apparent melting temperatures deduced from the cooling curves. All the curves show a minimum located around 60 or 65 wt.% zinc stearate. However, there is a major difference in that the apparent melting temperatures, inferred from the halts in the cooling curves, are significantly lower that the fluidification onset temperatures obtained from the rheological studies.



Figure 8. Fluidification temperatures from rheology measurements compared to apparent melting temperatures obtained from cooling curves.

The minima in the apparent melting- and fluidification temperatures are reminiscent of eutectic formation. However, the XRD diffraction results showed that the metal stearates co-crystallised. Conventional co-crystallisation leads to the formation of a complex with a fixed

composition ratio for the constituents. The melting curves in the phase diagrams of such systems are usually W-shaped corresponding to the formation of two eutectics, each with the complex participating with one of the parent compounds [12]. This feature is not shown by the present data. This is attributed to the fact that co-crystallisation occurred over a wide range of compositions, i.e. it led to the formation of solid solutions.

Optical microscopy results, obtained under polarised light, are presented in more detail in the Supplementary material. It discusses the microstructure evolution in the melts when cooling from 180 °C to ambient. Figure 9 shows micrographs for the solidified neat zinc stearate and a blend containing 35 wt.% magnesium stearate. Figure 9a shows that, upon cooling the neat zinc stearate melt, nucleation lead to polycrystalline growth patterns of a spherical shape, i.e. spherulites. According to Gránásy, Pusztai, Tegze, Warren and Douglas [13], such spherulitic growth patterns arise "from the competition between the ordering effect of local crystallographic symmetries and the randomization of the local crystallographic orientation that accompanies crystal grain nucleation at the growth front". These spherulites grew to about 250 µm in size before they impinged on neighbouring spherulites. The calcium stearate showed very different behaviour. There was evidence for the gradual development of texturing almost to fine-scaled to discern with the available equipment. It is possible that the observed change in texturing was due to the nucleation and growth of countless crystals that were too small to be observable at the $20 \times$ magnification in the optical microscope. The complex thermotropic behaviour exhibited by magnesium stearate [10] was not resolved in the present optical microscopy study and the cooled solid was featureless.

Figure 9b shows a micrograph recorded for the 35/65 magnesium/zinc stearate mixture. The granulated microstructure shows strong local orientation under polarised light. It probably arose from extensive nucleation that resulted in spherulites less than 50 μ m in size. The smaller size of the spherulites implies that there is also more interfacial area between them where there is a mismatch in crystalline orientations. It can be speculated that this might have contributed to the lower fluidification temperature of this blend.



Figure 9. Optical micrographs, under polarised light, of the solids obtained after cooling the melts of (a) neat zinc stearate and (b) the 35/65 magnesium/zinc stearate blend.



Figure 10. The effect of the metal stearate lubricants on the complex viscosity of 30 wt.% calcium carbonate compounds. Sasol F-T wax was added to the polyethylene matrix in order to lower the melt viscosity of the compounds.

The rheology data presented above indicates that mixtures of zinc stearate with either magnesium- or calcium stearate feature a lower fluidification temperature than the latter two lubricants. In theory, this should assist early wetting of a filler component, break-up of particle agglomerates and dispersion of individual particles into the matrix during a compounding process. Ultimately this should lead to filled compounds with a lower apparent viscosity. In order to test this hypothesis, 30 wt.% filled calcium carbonate compounds were prepared. Figure 10 shows rheology results obtained at 170 °C for a range of compounds in which the natures of both the wax component and the lubricants were varied. The measured viscosities were similar irrespective of the wax employed but showed significant variation depending on the nature of the metal stearate employed. All the compounds showed shear thinning behaviour over the frequency range 2 to 200 rad·s⁻¹. Those compounds based on neat calcium- or magnesium stearate feature a significantly higher complex viscosity than the compounds based on neat zinc stearate. Cutting back the zinc stearate, by replacing a one-third portion with one of the other metal stearates, had only a minor effect on the melt viscosity.

Figure 11 shows representative images that provide qualitative information about the distribution of the calcium carbonate particles near the surface of the blown films prepared by adding either neat magnesium stearate (Figure 11a) or a blend of magnesium stearate with zinc stearate (Figure 11b). Particle agglomerates were present in all the film samples. In Figure 11a there are ten large red spots representing particle agglomerates. Two of them are quite large. In Figure 11b only six such spots are visible and they appear to be smaller in size. In addition, there are many smaller red dots in Figure 11b that are of a size expected for the individual calcium carbonate particles.



Figure 11. EDS visualisation of the dispersion of calcium carbonate particles near the polymer film surfaces. (a) Magnesium stearate as lubricant, and (b) a blend of magnesium- and zinc stearate as lubricants. The smallest dots indicate the presence of primary calcium carbonate particles and the larger red spots are due to large-scale particle agglomerates.

Figure 12 shows a plot of the ImageJ results. It shows that in the sample based on magnesium stearate as the lubricant the particle sizes extended over a wide range, with large clumps measuring up to 13 μ m in diameter. This is reflected in the larger mean Feret diameter and larger standard deviation, i.e. $2.69 \pm 2.42 \mu$ m. The corresponding values for the film made using the metal stearate blend were $2.08 \pm 1.19 \mu$ m. The F-test and the t-test returned highly significant alpha values, confirming significant (*p* > 0.001) differences between these populations. These observations suggest that the metal stearate blend was more effective at breaking down agglomerates and facilitating the dispersion of individual CaCO₃ particles in the polymer matrix. The implication is that the reduced melt viscosities, observed for the zinc stearate as lubricant, can in part, be attributed to improved particle dispersion in the composites. While these preliminary results are encouraging, more work is required to confirm and quantify any advantage of using blends of zinc stearate with either calcium or

magnesium stearate as a lubricant package in filled polymer compounds. In particular, the performance in compounds containing much higher filler levels should be investigated. Such efforts are currently under way.



Figure 12. Particle Feret size distribution inferred from ImageJ analysis of the images shown in Figure 11.

4 CONCLUSIONS

Binary mixtures of zinc stearate with magnesium stearate and calcium stearate were investigated as possible alternative lubricants for polymer processing. It was hypothesised that these systems would form low melting eutectics which would improve filler dispersion. However, X-ray diffraction results indicated that both systems formed co-crystallised forming solids solutions with variable compositions as opposed to eutectic mixtures. In the calcium/zinc stearate system, it was observed that the zinc ions form part of the calcium stearate crystal lattice even at compositions of only 20 wt.% calcium stearate. The magnesium stearate system behaved differently. The low magnesium stearate compositions showed an increasing d-spacing as the magnesium stearate content was increased. This continued up to 50 wt.% magnesium stearate. At even higher magnesium stearate content, two sets of reflections were identified, indicating the presence of two phases for the dehydrated stearates. No significant difference was observed for calcium stearate and zinc stearate. A major shift to higher reflections was identified for the magnesium stearate.

The mixtures of lubricants were investigated via DSC and cooling curves. They showed lower melting points compared to the pure components. The lowest being about 66 °C for the 40 wt.% magnesium stearate sample and about 82 °C for 40 wt.% calcium stearate sample. Multiple endothermic peaks and shoulders were identified for all the mixtures. Some of the peaks can probably be attributed to a rotator phase transition of the mesophase.

The lowering of the melting point of these mixtures also resulted in a lower melt viscosity. Rheology of the calcium stearate mixtures showed reduced fluidification temperatures for all of the samples. The magnesium stearate system behaved differently, where samples behaved as either magnesium stearate dominant, zinc stearate dominant, or what is believed to be the cocrystal structure dominated system. Of the latter, two samples showed reduced fluidification temperatures compared to the pure zinc stearate.

These results would indicate the possibility of improved lubrication and dispersion of fillers for masterbatch compositions. An example of this was shown where 1:2 weight ratio of zinc stearate to magnesium or calcium stearate gave similar complex viscosity measurements to that of the industry standard zinc stearate.

CRediT authorship contribution statement

Justin Phillips: Conceptualization, Data curation, Formal analysis, Investigation, Methodology, Project administration, Software, Supervision, Validation, Visualization, Writing - original draft, Writing - review & editing. Michelle Weldhagen: Data curation, Formal analysis, Investigation, Methodology, Writing - original draft. Thobile Mhlabeni: Conceptualization, Data curation, Formal analysis, Investigation, Methodology, Resources, Software, Supervision, Writing - original draft. Lucky Radebe: Data curation, Formal analysis, Investigation, Methodology, Project administration, Writing - original draft. Shatish Ramjee: Conceptualization, Data curation, Formal analysis, Investigation, Methodology, Resources, Supervision, Validation, Visualization, Writing - original draft, Writing - review & editing. James Wesley-Smith: Conceptualization, Data curation, Formal analysis, Investigation, Methodology, Resources, Software, Validation, Visualization, Writing - review & editing. Maria Atanasova: Conceptualization, Data curation, Formal analysis, Investigation, Methodology, Resources, Software, Validation, Visualization, Writing original draft, Writing - review & editing. Walter W. Focke: Conceptualization, Formal analysis, Funding acquisition, Investigation, Methodology, Project administration, Resources, Supervision, Validation, Visualization.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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